and  $\frac{1}{2}\frac{1}{2}\frac{1}{2}$  cannot have more than about one-tenth of the contribution of free  $O_2$  molecules (Van Vleck, 1932) to the magnetic susceptibility of  $\beta$ - $O_2$ . Magnetic and crystallographic studies of dilute solutions of  $O_2$  in  $\beta$ - $F_2$  or in  $\beta$ - $N_2$  would be of interest in elucidating the disorder and interactions.

Table 1. Structure factors for  $\beta$ -F<sub>2</sub> and  $\gamma$ -O<sub>2</sub>

	β-	$eta$ – $\mathbf{F_2}$		$\gamma$ – $\mathrm{O_2}$		
hkl	$\overline{F_o}$	$F_c$	$\overline{F_o}$	$F_c$		
110		1.8	_	1.8		
200	20.3	17.4	34.7	34.8		
210	$23 \cdot 2$	$22 \cdot 9$	$37 \cdot 1$	40.9		
211	$22 \cdot 2$	21.2	36.8	38.4		
220	5.4	-5.7	$9 \cdot 9$	-7.5		
310	$2 \cdot 0$	$2 \cdot 2$	4.9	$3 \cdot 3$		
222	6.0	-2.6	17.9	-13.5		
320	$3 \cdot 4$	-1.6	13.9	-9.9		
321	$3 \cdot 7$	$3 \cdot 5$	13.7	12.4		
400	$7 \cdot 4$	$7 \cdot 2$	20.9	22.7		
410	6.0	-5.5	$12 \cdot 2$	-10.3		
411	$3 \cdot 1$	$-4 \cdot 1$	$7 \cdot 3$	-7.2		
330	$3 \cdot 6$	$3 \cdot 2$	$8 \cdot 4$	$5 \cdot 3$		
420	$2 \cdot 2$	$1 \cdot 2$	10.6	7.4		
421	$2 \cdot 3$	1.9	$9 \cdot 4$	8.0		
<b>332</b>	1.7	1.1	$4 \cdot 4$	1.7		
<b>422</b>	3.5	$3 \cdot 1$	7.9	$7 \cdot 2$		
430	$2 \cdot 5$	-3.0	$6 \cdot 3$	-7.0		
510		-0.8		0.1		
431	1.7	-0.5	$4 \cdot 2$	-1.8		
520	1.9	0.8	$4 \cdot 2$	1.6		
<b>432</b>	$2 \cdot 0$	1.7		1.3		
521	$1 \cdot 3$	-0.8		-0.5		
440	2.5	-1.5		-0.9		
530	1.3	0.9	_	$2 \cdot 9$		
433		0.7		1.2		
531	_	0.0		0.0		
600	2.1	$-2\cdot 1$	7.8	-7.3		
442	$2 \cdot 1$	0.3	1.7	2.8		
610		1.3		4.4		
611		0.1	4.4	2.5		
$\begin{array}{c} 532 \\ 620 \end{array}$	1.7	-1.0	4.3	-2.7		
620	1.9	- 1·8	4·1	-6.0		
		0.4	$4 \cdot 3$	-3.2		
540	_	1.0		3.6		
541	_	0.1		0.1		
622		-0.2		-0.6		
630		$-0.2 \\ -1.0$		-0.4		
$\begin{array}{c} 542 \\ 631 \end{array}$	_		_	-4.0		
631 444	-	$-0.1 \\ -1.7$	4.3	0.6		
632	$2 \cdot 6$		4.1	-6.3		
032		0.3	$2 \cdot 9$	2.4		

The agreement obtained so far is about equally good for (a) spherically disordered  $O_2$  (or  $F_2$ ) at 000 and  $\frac{1}{2}\frac{1}{2}\frac{1}{2}$  and cylindrically disordered  $O_2$  in the other sites so that the bond lies in the plane of disorder  $(R=0\cdot16)$ , and (b) a statistical model in which the F atoms of the molecular twofold site are in 16-fold positions ( $\frac{1}{4}$  F's on threefold axes), and the F atoms of the other molecules are in 24-fold positions ( $\frac{1}{2}$  F's in the plane of disorder). Large isotropic temperature factors are also required for model (b), which also yields  $R=0\cdot16$  for either  $\gamma\cdot O_2$  or  $\beta\cdot F_2$  (Table 1). Three-dimensional electron density maps also indicate spherical disorder for the  $O_2$ 's  $(F_2$ 's) at 000 and  $\frac{1}{2}\frac{1}{2}\frac{1}{2}$ , but oblate spheroidal distributions for the other  $O_2$ 's. Further studies of the many related types of disorder are in progress.

Note added in proof.— Later refinements have reduced R to 0·10 for  $\beta$ -F<sub>2</sub> and 0·13 for  $\gamma$ -O<sub>2</sub>.

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On the symmetry of phases in the reciprocal lattice: a simple method. By E. F. Bertaut, Laboratoire d'Electrostatique et de Physique du Métal, Institut Fourier, Grenoble, France

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No mention is made in recent papers on reciprocal lattice symmetry (Bienenstock & Ewald, 1962; Jeffery, 1963) of the technique of structure factor algebra to find phase relations between structure factors. Our simple 'do-it-yourself method' uses the information already contained in *International Tables for X-ray Crystallography* (1952).

We only require the knowledge of the so called general positions which is equivalent to the knowledge of the symmetry elements. Indeed to every symmetry operation  $\mathbf{r} \rightarrow \mathbf{r}'$  corresponds a phase relation in reciprocal space.

Let us illustrate the structure factor algebra method by the example of space group  $P4_322(D_4^2)$  No. 95 where

the general position is  $\pm(x, y, z; \overline{x}, \overline{y}, \frac{1}{2} + z; \overline{y}, x, \frac{3}{4} + z; y, \overline{x}, \frac{1}{4} + z)$  and let us find the phase relation corresponding to the symmetry operation  $\mathbf{r}(xyz) \rightarrow \mathbf{r}'(\overline{y}, x, \frac{3}{4} + z)$ .

This is done in four steps.

- (1) Form the scalar product  $\mathbf{h} \cdot \mathbf{r}' = h(\overline{y}) + kx + l(\frac{3}{4} + z)$ .
- (2) Isolate the translational component which gives rise to a phase factor  $a = \exp 2\pi i \frac{3}{4}l = \exp(3\pi i l/2)$ .
- (3) Rewrite the resting scalar product in the form  $\mathbf{r} \cdot \mathbf{h}' = xk + y(\hbar) + zl$ , *i.e.* bring xyz into their usual sequence.
- (4) The desired symmetry relation between structure factors is

 $F(\mathbf{h}) = aF(\mathbf{h}'), i.e.$  $F(hkl) = \exp(3\pi i l/2)F(k\overline{h}l).$ 

If the reader is able to show that to the points  $\mathbf{r}(xyz)$  and  $\mathbf{r}'(\frac{1}{4}-z,\frac{1}{4}-y,x)$  of group  $Fd3m(O_h^2)$  No. 227 corresponds the phase relation  $F(hkl)=F(l\bar{k}h)\exp{(\pi i(h+k)/2)}$  he will probably have understood how to use the method.

The theoretical proof may be found in French (Bertaut, 1959), in English (Bertaut & Waser, 1957) and in German (Bertaut, 1958).

In Jeffery's (1963) notation, the phase relation under (4) reads

 $\alpha_{hkl} = \alpha_{k\bar{h}l} + (3\pi/2) \, l$ 

whereas Jeffery writes

$$\alpha_{hkl} = \alpha_{khl} + (3\pi/2) l$$
.

We must conclude and have checked that in Tables 1 and 2 given by Jeffery,  $\alpha_{\bar{k}hl}$  should be replaced by  $\alpha_{\bar{k}\bar{h}l}$  and, vice versa,  $\alpha_{k\bar{h}l}$  should be replaced by  $\alpha_{\bar{k}hl}$ .

Finally let us mention that all phase relations for centrosymmetric groups are given in *Tables de Linéarisation des Facteurs de Structure* (Bertaut & Dulac, 1955, 1956).

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Lattice constants of some double sulphates  $M_2Co(SO_4)_2$ .  $6H_2O$ . By P. Hartman and C. F. Woensdregt, Geologisch en Mineralogisch Instituut der Rijksuniversiteit, Garenmarkt 1b, Leiden, The Netherlands

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The lattice constants of four isomorphous cobalt double sulphates  $\rm M_2Co(SO_4)_26H_2O$ , with  $\rm M=K$ , Rb, NH<sub>4</sub> and Cs, have been determined from powder X-ray photographs with a quadruple Guinier—de Wolff camera (de Wolff, 1948) and Cu  $\rm K\alpha$  radiation. Because of the larger deviations on the two outer strips, only the inner ones were used, one recording the powder lines of the cobalt double sulphate, the other recording the lines of a reference substance for which we chose potassium aluminum alum. In a separate measurement on a Debye-Scherrer X-ray photograph taken in a cylindrical camera with 9 cm diameter the unit cell edge of the potassium aluminum alum was found to be  $12\cdot157_8$  Å, in excellent agreement with the value  $12\cdot158\pm0\cdot001$  Å found by Klug & Alexander (1940).

Calibration of each film was based on this value of the

alum unit cell edge. The measured line positions appeared to be, within the limits of errors, a linear function of the calculated positions.

Irregular deviations as found by Fisher (1957) were not observed.

With this linear function the line positions of the sulphates were corrected and the constants in the relation

$$\sin^2\theta = Ah^2 + Bk^2 + Cl^2 + Dhl$$

were calculated. B was found from  $\sin^2\theta$  differences between  $hk_1l$  and  $hk_2l$  reflexions, A from hk0 reflexions, C from 0kl reflexions and from  $\sin^2\theta$  differences between hkl and hkl reflexions and D from all hkl reflexions. Table 1 gives the calculated lattice constants with the standard deviation of the mean. The morphological axial ratios were taken from Groth (1908).

Table 1. Lattice constants of M<sub>2</sub>Co(SO<sub>4</sub>)<sub>2</sub>.6H<sub>2</sub>O

	$\mathbf{K}$	${f R}{f b}$	$\mathbf{NH_4}$	Cs
$egin{array}{c} a_{f 0} \ b_{f 0} \ c_{f 0} \end{array}$	$\begin{array}{c} 9.061 \pm 0.003 \text{ Å} \\ 12.207 \pm 0.001 \\ 6.151 \pm 0.002 \end{array}$	$\begin{array}{c} 9 \cdot 180 \pm 0 \cdot 001 \text{ Å} \\ 12 \cdot 433 \pm 0 \cdot 001 \\ 6 \cdot 230 \pm 0 \cdot 003 \end{array}$	$\begin{array}{c} 9 \cdot 247 \pm 0 \cdot 001 \text{ Å} \\ 12 \cdot 519 \pm 0 \cdot 001 \\ 6 \cdot 239 \pm 0 \cdot 001 \end{array}$	$\begin{array}{c} 9.316 \pm 0.001 \text{ Å} \\ 12.824 \pm 0.001 \\ 6.365 \pm 0.002 \end{array}$
$eta \ eta$ (morph.)	104° 48′±2′	106° 01′±1′	$107^{\circ} 02' \pm 1'$	107° 07′±1′
	104° 55′	106° 01′	$106^{\circ} 56'$	107° 08′
Axial ratio	0.7423:1:0.5039	0.7384:1:0.5011	0.7386:1:0.4984	0.7265:1:0.4963
Axial ratio (morph.)	0.7404:1:0.5037	0.7391:1:0.5011	0.7392:1:0.4985	0.7270:1:0.4968